

Article

Photovoltaic Performance of $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ Perovskite Solar Cell

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Abstract: Perovskite solar cells have attracted extensive research attention recently as they are promising high-performance solar cells with long-time stability at a low cost. In this study, we demonstrated a one-step solution approach to prepare the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer by adding lead chloride (PbCl_2) to the standard methylamine iodide (MAI) precursor solution and annealing process of the perovskite layer at different temperature. Finally, the Ag/Spiro-OMeTAD/ $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ /mp-TiO₂/c-TiO₂/FTO/Glass perovskite solar cell was successfully fabricated by using solution processing.

Keywords: $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$, TiO₂, Spiro-MeOTAD, Perovskite solar cell

1. Introduction

Global energy requirement has been continually increasing with industrial development and population growth in recent years. To date, over 85% of energy attrition is generated from fossil fuels, which causes environmental pollution and climate change. In this context, the solar cell is one of the most promising technologies that directly convert sunlight into electricity. Solar energy possesses cost-effective, reliable, and safe attributes [1,2].

The first solar cell was fabricated by using the silicon p-n junction. The silicon solar cells are currently leading the commercial photovoltaics markets with a share of 85%. The best power conversion efficiency (PCE) of the silicon solar cell is about 28% (NREL: Best research cell efficiency). In recent years, perovskite solar cells (PSCs) have received extensive attention owing to their potential for achieving lightweight, cheap, and facile fabricated solar cells [3]. The perovskite solar cell is a semiconductor with a special crystal structure that makes them well suited for solar cell technology and can be manufactured at room temperature with much less energy and more sustainability to produce than the silicon solar cell.

The organic-inorganic hybrid metal halide perovskite has a crystal structure with the general formula ABX_3 composed of small organic cations (A^+), metal cations (B^+), and halide anions (X^-). Methylammonium lead ($\text{CH}_3\text{NH}_3\text{PbX}_3$, $\text{X} = \text{I}, \text{Br}, \text{or Cl}$) halide perovskites represent a novel class of absorbers for solar conversion applications [4]. Various perovskite materials (e.g., $\text{CH}_3\text{NH}_3\text{PbI}_3$, $\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Cl}_x$, and $\text{CH}_3\text{NH}_3\text{PbBr}_3$) and device composition (e.g., mesoporous and planar structures) have been investigated with promising results by using solution processing, thermal evaporation, or sputtering method [5–7]. The $\text{CH}_3\text{NH}_3\text{PbBr}_3$ perovskite solar cells have attracted less attention than the iodide homologous. The relatively wide band gap of 2.3 eV limits the usable spectral range for single junction solar cells [8]. Most impressive, the $\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Cl}_x$ cells exhibit high V_{oc} values and reach qV_{oc}/E_{Gap} values of up to 0.71 and a remarkably high figure for a new type of cell [9]. In contrast to the success of $\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Cl}_x$ for planar devices, the solution process of $\text{CH}_3\text{NH}_3\text{PbI}_3$ has mainly been used in mesoporous cell structures [10].

The improvement of the PCE of PSCs is dependent on how to control the shape, interface defects, and passivation of perovskite thin films [11]. In this study, methylamine iodide (MAI) and lead chloride (PbCl_2) mixed with dimethylformamide (DMF) were used as the perovskite solutions for the fabrication of the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer in the one-step spin-coating method. The different annealing temperatures were used to understand the effect on the perovskite layer. Finally, the Ag/Spiro-OMeTAD/ $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ /mp-TiO₂/c-TiO₂/FTO/Glass structure was fabricated as the perovskite solar cell.

2. Materials and Methods

2.1. Materials

All chemicals have been used without further purification. $\text{CH}_3\text{NH}_3\text{I}$ (MAI, Sigma-Aldrich, 99.5%), Lead(II) chloride (PbCl_2 , Sigma-Aldrich, 99.99%), Titanium isopropoxide (TTIP, Sigma-Aldrich, 97%), Titanium(IV) oxide (TiO₂, Sigma-Aldrich, 99.5%), Spiro-OMeTAD (Lumtec), Li-TFSI (Sigma-Aldrich), tert-butylpyridine (tBP, Sigma-Aldrich, 98%), chlorobenzene (Sigma-Aldrich,

99.9%), Dimethylformamide (DMF, Sigma-Aldrich) and silver target (Ag, Parekh Industries, 99.9%) were used for the perovskite solar cell device fabrication.

2.2. Fabrication of Solar Cell

Firstly, patterned FTO glass substrates were etched by using Zn powder and HCl. Then, the FTO substrates were ultrasonically cleaned with Hellmanex (c) III, deionized water, acetone, and isopropanol sequence. Afterward, they were dried with high-pure N₂ gas. Finally, the substrates were treated with UV-O₃ for 15 min. The surface SEM image of the FTO is shown in Fig. 1. The compact TiO₂ (c-TiO₂) and mesoporous TiO₂ (mp-TiO₂) as the electron-transport layer (ETL) deposited on the FTO substrate. The c-TiO₂ was prepared by the TTIP and diluted in ethanol as a precursor. The TTIP precursor was coated by spin coating method with a speed of 4000 rpm for 30 s and then dried at 150 °C for 1 h. Subsequently, ethanol dispersed mp-TiO₂ nanoparticles were spin-coated on the c-TiO₂ and dried at 600 °C for 1 h. The CH₃NH₃PbI₂Cl perovskite solution, the MAI, and PbCl₂ were dissolved in pure DMF solvent and prepared in a glove box. The CH₃NH₃PbI₂Cl solution was stirred for 2 h and filtered before use. For the solar cell device fabrication, the CH₃NH₃PbI₂Cl perovskite solution was spin-coated on the c-TiO₂/FTO/Glass substrate at 2000 rpm for 30 s, and the substrate was annealed at 110 °C for 10 min. When the CH₃NH₃PbI₂Cl /c-TiO₂/FTO/Glass substrate cooled down to room temperature, the spiro-OMeTAD solution as the hole-transport layer (HTL) was spin-coated on the CH₃NH₃PbI₂Cl perovskite layer at 4000 rpm for 60 s. The samples were kept in a drier bin overnight. The spiro-OMeTAD solution was composed of spiro-OMeTAD, tBP, and Li-TFSI solution in chlorobenzene. Finally, 100 nm-thick silver (Ag) electrodes were deposited in a sputtering system. The effective active area of the solar cell device was 0.04 cm².

2.3. Measurement Equipment

The high-resolution scanning electron microscopy (HRSEM, SU8000, Hitachi Company) at 20 kV was used to observe the materials and devices' structure. The crystallography of the materials was measured through X-ray diffraction (XRD, D8 Discover, Bruker Company) with CuK α radiation ($2\theta = 20\text{--}60^\circ$). The absorption of the materials was measured by using an ultraviolet-visible (UV-vis) spectrophotometer (Analytik Jena). The current-voltage (I-V) curve of perovskite solar cells was measured by using Keithley 2450 source measuring unit under solar-simulated light (Class AAA Solar Simulators). The 100 mW/cm² (AM 1.5G) sunlight was calibrated with a standard Si-based solar cell.

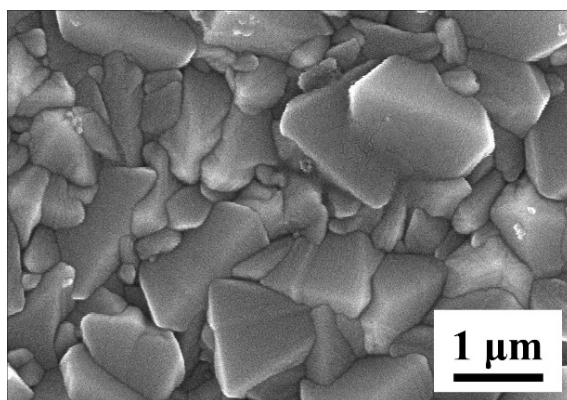


Fig. 1. SEM image of the FTO/Glass substrate.

3. Results and Discussion

To investigate the morphology and thickness of the c-TiO₂ layer, scanning electron microscope (SEM) measurements were performed. Figure 2 shows that the c-TiO₂ layer deposited on the FTO substrate has a mirror-like smooth surface morphology and the thickness of the c-TiO₂ layer of about 20 nm. Because the mp-TiO₂ layer has a larger grain size than the c-TiO₂ layer, a less dense film is obtained. Therefore, the c-TiO₂ layer efficiently reduces the carrier recombination induced by the direct contact between the CH₃NH₃PbI₂Cl perovskite layer and the FTO substrate. The mp-TiO₂ nanoparticles are dispersed in different ethanol contents as shown in SEM images (Fig. 2). The ratio of the TiO₂ nanoparticles and ethanol is equal to 1:2, and the thickness of the mp-TiO₂ layer is 432 nm (Fig. 2(a)). When the ratio of the TiO₂ nanoparticles and ethanol is equal to 1:4, the thickness of the mp-TiO₂ layer is 257 nm (Fig. 2(b)). The thickness of the mp-TiO₂ layer decreases as the ethanol content increases. In Fig. 2(c), it is

found that the mp-TiO₂ layer has a large grain size than the c-TiO₂ layer, and the nanopores exist on the surface of the c-TiO₂ layer. In the mp-TiO₂/c-TiO₂/FTO/Glass structure, the pores act as nucleating sites for the CH₃NH₃PbI₂Cl grain growth.

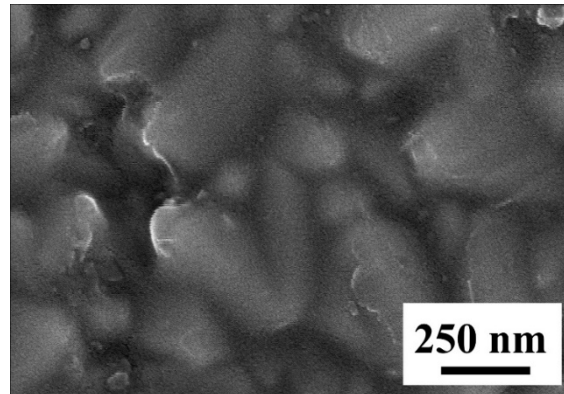


Fig. 2. SEM image of the compact TiO₂ deposited on FTO/glass substrate.

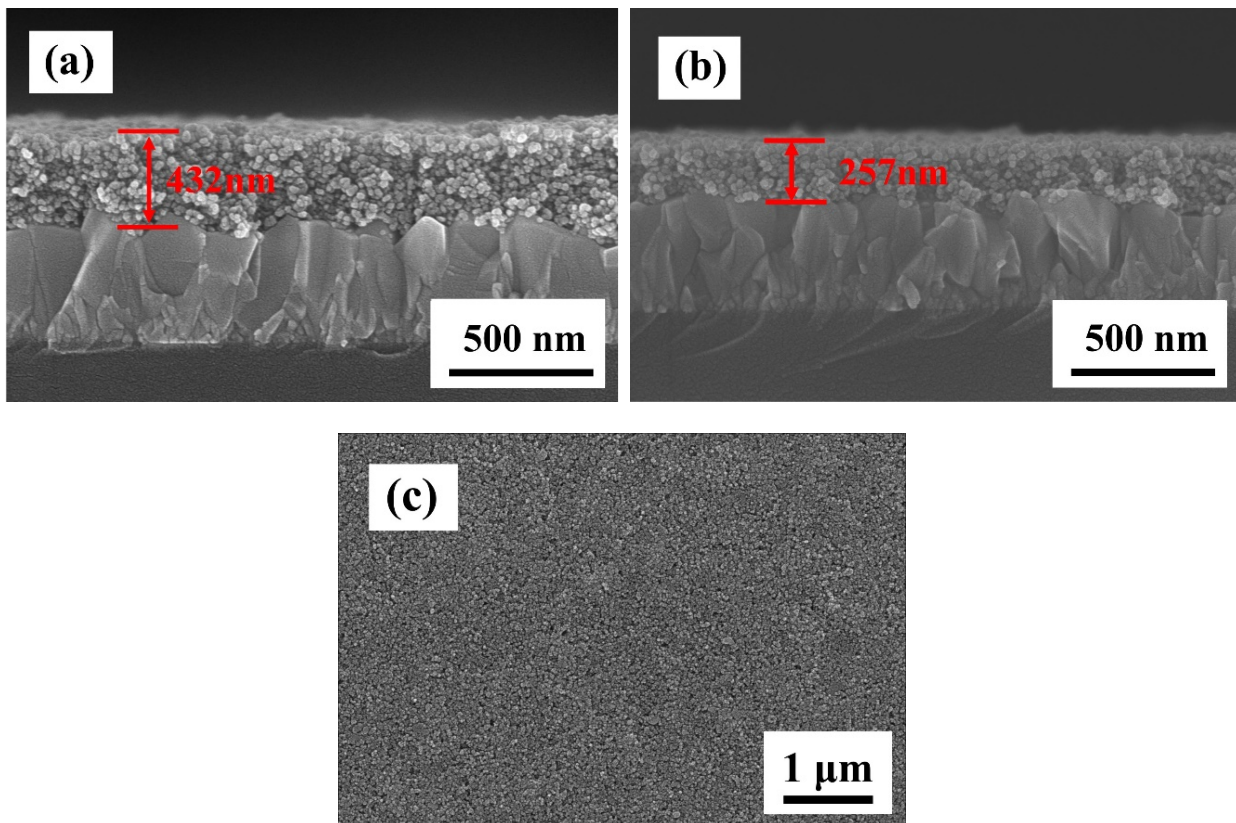


Fig. 3. Cross-section SEM image of mesoporous TiO₂ layer as the ratio of the TiO₂ nanoparticles and ethanol equal to (a)1:2 and (b)1:4. (c) The surface SEM image of the mesoporous TiO₂ layer.

Figure 4 shows the XRD pattern of the mp-TiO₂ layer. At the annealing temperature of 600 °C, the sharp peaks in the diffraction pattern are shown, which indicates that the high crystallinity of TiO₂ powder is obtained, and the diffraction peaks have good consistency with anatase TiO₂ and rutile TiO₂ (JCPDS 21-1272 and JCPDS 75-1537). No second phases or unknown phases exist in Fig. 4. This result demonstrated that the mp-TiO₂ is appropriate for application on ETL.

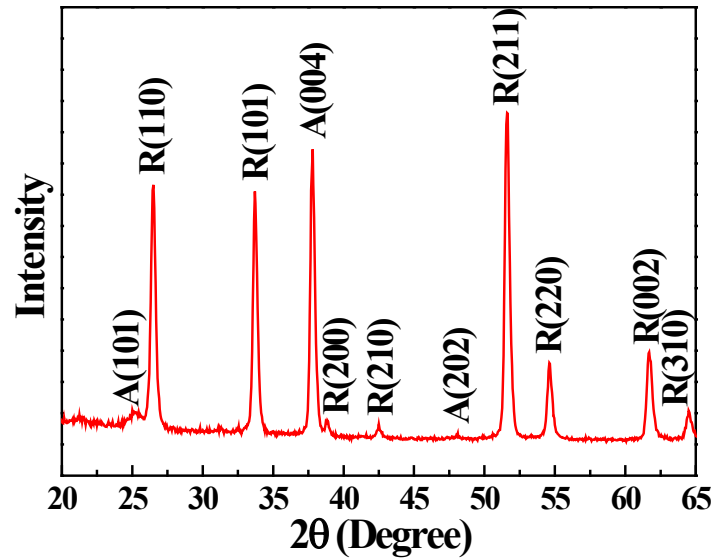


Fig. 4. XRD pattern of mesoporous TiO₂.

Figure 5 shows the SEM image of the surface of the CH₃NH₃PbI₂Cl perovskite layer. At the annealing temperature of 80 °C, the large elongated crystal is formed with a significant portion of the FTO substrate that is exposed without CH₃NH₃PbI₂Cl perovskite material coverage. When annealed at 110 °C, the elongated large crystal disappears, and the substrate is coated with relatively small crystals that are interconnected with a high surface coverage. The perovskite film is full of compact nanocrystals without pinholes, which makes the film smoother than sequentially deposited ones as shown in Fig. 5(b). The difference in the surface coverage of perovskite films affects the device characteristics.

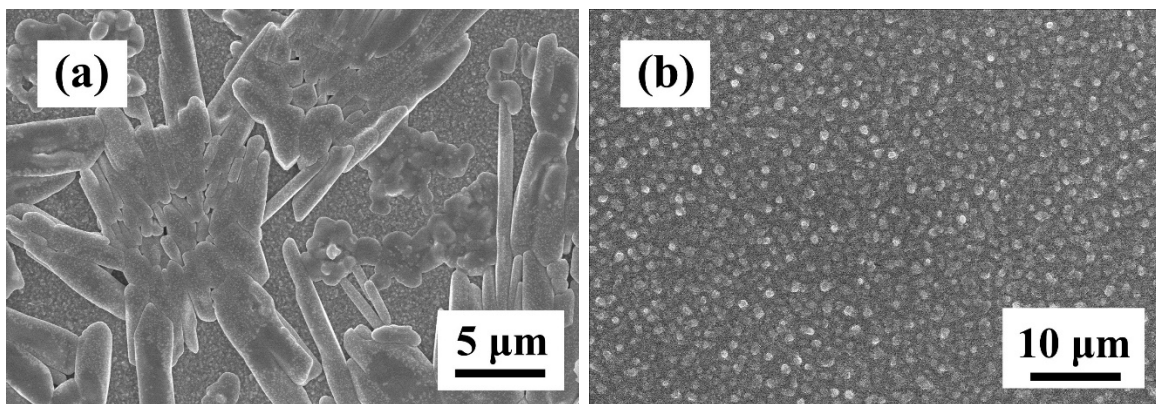


Fig. 5. SEM image of the surface of the CH₃NH₃PbI₂Cl perovskite layer as the annealing temperature at (a) 80 °C and (b) 110 °C.

The X-ray diffraction (XRD) patterns of the CH₃NH₃PbI₂Cl deposited on the FTO substrate as shown in Fig. 6. The diffraction peaks of the CH₃NH₃PbI₂Cl perovskite material at 23.1°, 28.3°, 35.2°, 40.4°, 43.1°, 48.2°, 53.9°, and 55.1° are apart from the three peaks from the FTO substrate (marked as o) [12,13]. No PbCl₂ phases or unknown phases exist in Fig. 6. Figure 7 shows the UV-vis absorption spectra of the CH₃NH₃PbI₂Cl perovskite layer at the annealing temperature of 110 °C. When the UV-vis light of different wavelengths irradiates the sample continuously, the absorption intensity corresponding to the wavelength is obtained. Figure 7 shows that the CH₃NH₃PbI₂Cl perovskite layer has a strong excitonic absorption edge at 800 nm.

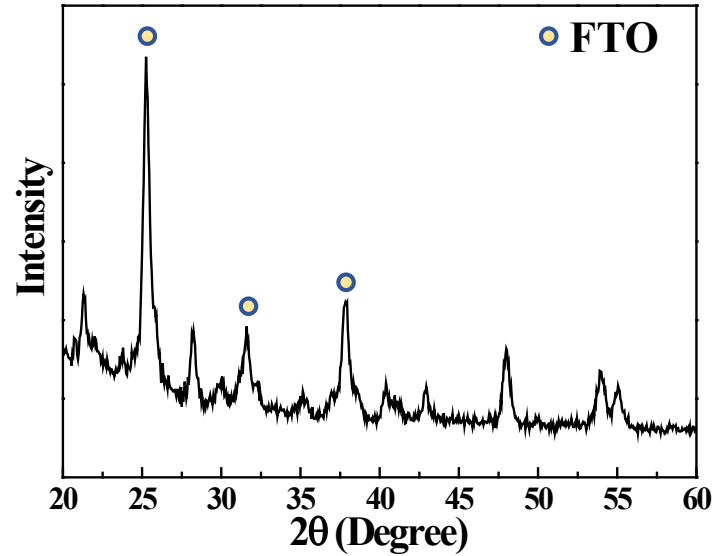


Fig. 6. XRD pattern of CH₃NH₃PbI₂Cl perovskite layer at the annealing temperature of 110 °C.

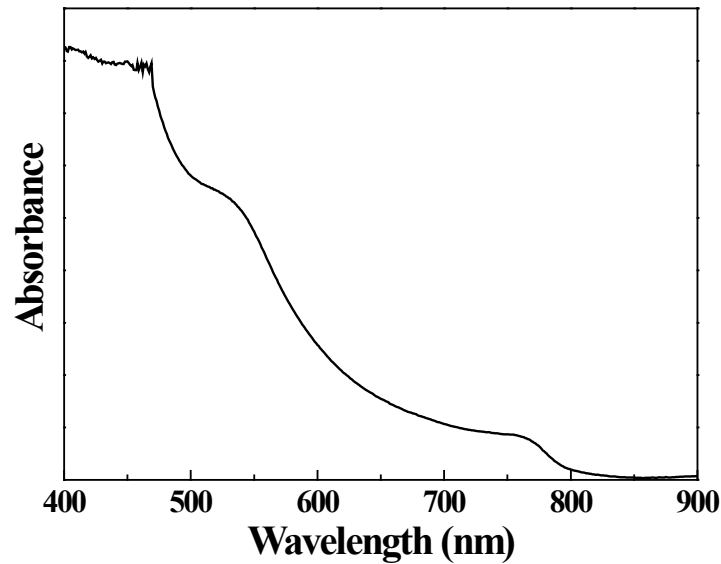


Fig. 7. UV-vis absorption spectra of CH₃NH₃PbI₂Cl perovskite layer at the annealing temperature of 110 °C.

The overall power conversion efficiency (η) of the solar cell is calculated from the short-circuit photocurrent density (J_{sc}), open-circuit photovoltage (V_{oc}), fill factor (F.F.) of the cell, and the incident light intensity ($P_{in} = 100 \text{ mW/cm}^2$). Therefore, the more square-like current density-voltage (J-V) curve of the CH₃NH₃PbI₂Cl perovskite solar cell is essential for achieving the maximum value of F.F. (Fig. 8). η and F.F. values are calculated by using the following equation.

$$\eta = \frac{P_{max}}{E} = \frac{J_{sc} \times V_{oc} \times F.F.}{P_{in}} \quad (1)$$

$$F.F. = \frac{I_{max} \times V_{max}}{J_{sc} \times V_{oc}} \times 100\% \quad (2)$$

The perovskite solar cell demonstrates a J_{sc} of 0.03 mA/cm^2 , V_{oc} of 0.81 V , and F.F. of 0.641% to yield a η of 0.21% . Figure 9 shows the cross-section SEM image of the Spiro-OMeTAD/ $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ /mp-TiO₂/c-TiO₂/FTO/Glass structure. The thickness of the c-TiO₂, mp-TiO₂, $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$, and Spiro-OMeTAD are 34 , 245 , 200 , and 169 nm , respectively. In Fig. 9, it found that pores exist in the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer, therefore, the η of the perovskite solar cell presents a lower value.

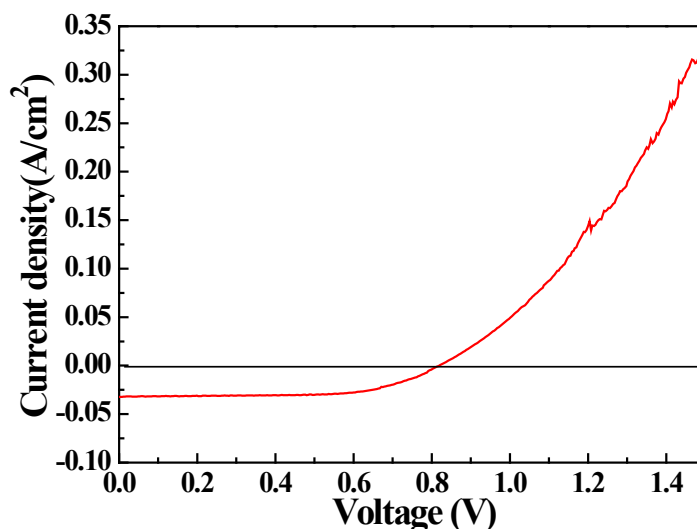


Fig. 8. I-V curves of the Spiro-OMeTAD/ $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ /mp-TiO₂/c-TiO₂/FTO/Glass perovskite solar cell.

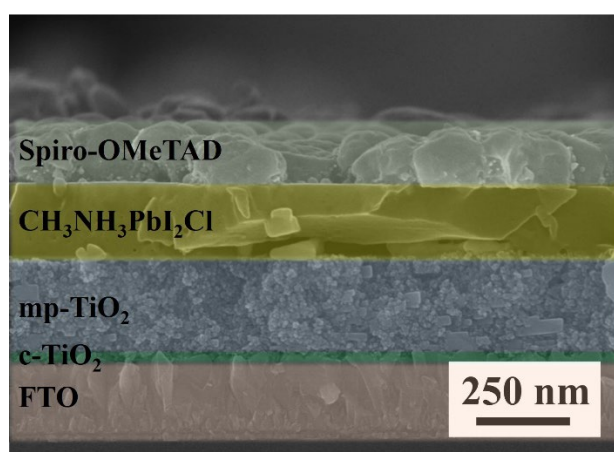


Fig. 9. The cross-section SEM image of the Spiro-OMeTAD/ $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ /mp-TiO₂/c-TiO₂/FTO/Glass structure.

5. Conclusions

In summary, we prepare perovskite $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ on a mesoporous mp-TiO₂/c-TiO₂/FTO/Glass substrate with a one-step solution. From the SEM, XRD, and UV-vis analysis, it was found that the optimal annealing temperature of the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer is set to $110 \text{ }^\circ\text{C}$. The Spiro-OMeTAD as the HTL deposition on the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer and the $\text{CH}_3\text{NH}_3\text{PbI}_2\text{Cl}$ perovskite layer is kept in a desiccator overnight. Finally, the optimized perovskite solar cell was obtained with the J_{sc} of 0.03 mA/cm^2 , V_{oc} of 0.81 V , and F.F. of 0.641% , yielding a η value of 0.21% .

Author Contributions: The manuscript was written with the contributions of all authors. All authors have approved the final version of the manuscript. Yu-Ting Tseng performed the perovskite materials synthesis and analysis. Yi-Tzu Tseng performed the perovskite solar cell fabrication. Chia-Ching Wu (Corresponding Author) performed the design of this research, and analyzed the material and device, and wrote the article.

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Conflicts of Interest: The authors declare no conflict of interest.

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